



OPEN

Carbon dioxide and hydrogen adsorption study on surface-modified HKUST-1 with diamine/triamine

Tomas Zelenka¹, Klaudia Simanova², Robin Saini³, Gabriela Zelenkova¹, Satya Pal Nehra⁴, Anshu Sharma³ & Miroslav Almasi²✉

The present article intended to study the influence of post-synthetic modification with ethylenediamine (*en*, diamine) and diethylenetriamine (*deta*, triamine) within the coordinatively unsaturated sites (CUSs) of HKUST-1 on carbon dioxide and hydrogen storage. The as-synthesized adsorbent was solvent-exchanged and subsequently post-synthetically modified with di-/triamines as sources of amine-based sorption sites due to the increased CO₂ storage capacity. It is known that carbon dioxide molecules have a high affinity for amine groups, and moreover, the volume of amine molecules itself reduces the free pore volume in HKUST-1, which is the driving force for increasing the hydrogen storage capacity. Different concentrations of amines were used for modification of HKUST-1, through which materials with different molar ratios of HKUST-1 to amine: 1:0.05; 1:0.1; 1:0.25; 1:0.5; 1:0.75; 1:1; 1:1.5 were synthesized. Adsorption measurements of carbon dioxide at 0 °C up to 1 bar have shown that the compounds can adsorb large amounts of carbon dioxide. In general, *deta*-modified samples showed higher adsorbed amounts of CO₂ compared to *en*-modified materials, which can be explained by the higher number of amine groups within the *deta* molecule. With an increasing molar ratio of amines, there was a decrease in wt.% CO₂. The maximum storage capacity of CO₂ was 22.3 wt.% for HKUST-1: *en*/1:0.1 and 33.1 wt.% for HKUST-1: *deta*/1:0.05 at 0 °C and 1 bar. Hydrogen adsorption measurements showed the same trend as carbon dioxide, with the maximum H₂ adsorbed amounts being 1.82 wt.% for HKUST-1: *en*/1:0.1 and 2.28 wt.% for HKUST-1: *deta*/1:0.05 at – 196 °C and 1 bar.

Carbon dioxide emissions caused by anthropogenic activity, in particular electricity generation, industrial production and transport, account for about 80% of greenhouse gas emissions. For this reason, it is necessary to reflect and re-evaluate the currently used technologies and develop CO₂ adsorbents that will effectively capture CO₂ from the atmosphere. These changes will be possible to slow down, stop, and even reverse global warming and many unions and countries around the world are contributing to this goal. The European Union (EU) could be mentioned, which has committed all EU members to a 55% reduction in CO₂ emissions by 2030 and carbon neutrality by 2050¹. The U.S. The Department of Energy (DOE) has set an ambitious target of 90% capture of the CO₂ from natural gas flue streams² and many more.

Another way to reduce CO₂ emissions is to transform fossil fuel combustion transport as one of the primary producers of CO₂ into alternative energy sources. There are currently two options to choose: electric motors or hydrogen-powered engines. Although hydrogen engines' design, technology, and performance are at a high level, the problem is insufficient vehicle mileage. For this reason, there is a need to develop fuel tanks that can store large amounts of hydrogen. The U.S. Department of Energy (DOE) has set 2025 technical targets for hydrogen gravimetric capacity of 5.5 wt.% H₂, operating in the range of 40/85 °C³.

In both above-discussed applications, hybrid inorganic–organic materials called metal–organic frameworks (MOFs) are intensively studied and developed. MOFs belong to a class of coordination compounds composed

¹Department of Chemistry, Faculty of Science, University of Ostrava, 30. Dubna 22, 701 03 Ostrava, Czech Republic. ²Department of Inorganic Chemistry, Faculty of Science, P.J. Safarik University, Moyzesova 11, 040 01 Kosice, Slovak Republic. ³Department of Physics, School of Engineering & Technology, Central University of Haryana, Mahendergarh 123031, India. ⁴Center of Excellence for Energy and Environmental Studies, Deenbandhu Chhotu Ram University of Science and Technology, Murthal 131039, India. ✉email: miroslav.almasi@upjs.sk

of metal cations/clusters connected by organic linkers to form porous three-dimensional polymeric frameworks. Large surface area, different pore shapes/sizes, and pore volumes can be effectively tuned based on appropriately selected building blocks and their compatibility. Another advantage is the post-synthetic modifiability of frameworks which can increase the efficiency of the material in the selected application. MOFs find their usage in gas adsorption and separation^{4–7}, heterogeneous catalysis^{8,9}, sensors^{10–12}, magnetic refrigeration^{13,14}, drug delivery^{15,16}, or as SARS-CoV-2 detection and elimination materials^{17,18}. In CO₂ storage of unmodified MOF materials, the leaders at low pressures are: Mg-MOF-74 (30.1 wt.% @ 20 °C and 1 bar)¹⁹, Mg₂(dobpdc) (23.8 wt.% @ 40 °C and 1 bar)²⁰ and at high pressures UiO(bpdC) (79.7 wt.% @ 30 °C and 20 bar)¹⁹, NU-111 (61.8 wt.% @ 20 °C and 30 bar)²¹. Post-synthetic modification can increase the stored amount of CO₂²², such as tetraethylenepentamine-modified Mg-MOF-74 whose initial capacity of 23.4 wt.% has increased to 26.9 wt.% (@ 20 °C and 1 bar)²³, or polyethylenimine-modified MIL-101, 1.32 wt.% → 18.48 wt.% (@ 20 °C and 0.15 bar)²⁴. Hydrogen storage capacities in the top MOF materials @ – 196 °C and 1 bar ranged in 2.0–2.8 wt.%^{25,26} and at high pressures the best H₂ adsorbents include: DUT-32 (14.2 wt.% @ – 196 °C and 82 bar)²⁷, she-MOF-1 (12.6 wt. % @ – 196 °C and 100 bar)²⁸ and SNU-77H (11.0 wt.% @ – 196 °C and 90 bar)²⁹.

In the present study, we were inspired by amine-functionalized silicas with which our working group has many years of experience^{30,31}. We applied a similar approach for MOF material (HKUST-1), which was post-synthetically modified with diamine (ethylenediamine, *en*) and triamine (diethylenetriamine, *deta*) on coordinatively unsaturated sites (CUSs) within HKUST-1 framework. Amines are known to have a high ability to capture CO₂ molecules, and therefore HKUST-1 has been functionalized with mentioned amines in different molar ratios. It is known that the calculated ideal pore size of material for efficient H₂ storage is 6–7 Å^{32,33}. We modified HKUST-1 by varying the concentration of amines to achieve maximum H₂ storage. The prepared materials were subsequently studied as CO₂ and H₂ adsorbents at 0 °C and – 196 °C, respectively. It was shown that the enhanced capacity of the materials towards the selected adsorptives increased with an decreasing amine molar ratio. In addition, it can be concluded that better results were observed for *deta*-modified materials compared to *en*-functionalized materials.

Experimental

Chemicals, synthesis and post-synthetic modification. All chemicals used to synthesise HKUST-1 and its subsequent post-synthetic modification were purchased in the highest available purity from Acros Organics and Sigma Aldrich.

HKUST-1. 1 g (4.76 mmol) of benzene-1,3,5-tricarboxylic acid (H₃BTC) was dissolved in a 30 ml solvent mixture of ethanol and *N,N'*-dimethylformamide (DMF) (1:1, v:v). Subsequently, 15 ml of an aqueous solution of copper nitrate trihydrate (2.08 g, 8.6 mmol) was added to the H₃BTC solution. The mixture was stirred for 10 min and finally allowed to react in an oven at 85 °C for 24 h. After a mentioned time, the resulting product was filtered off and dried. The weight of the prepared HKUST-1 was 2.088 g. Further, DMF (b.p. 153 °C) molecules located in the cavities of as-synthesized (AS) HKUST-1 were exchanged (EX) for ethanol (b.p. 78 °C) for subsequent easier activation at lower temperatures to obtain HKUST-1 in activated form (AC). The solvent exchange process was performed in a Soxhlet extractor for 48 h.

Post-synthetic modification. 100 mg of activated HKUST-1 (200 °C, 30 min) was dispersed in 10 ml of dry methanol under a nitrogen atmosphere. A 0.5 M solution of amine (diethylamine (*en*), diethylenetriamine (*deta*)) in dry methanol was added to the suspension in various proportions. The selected molar ratios for Cu(II) in HKUST-1: amine were 1:2; 1:1.5; 1:1; 1:0.5; 1:0.25; 1:0.1 for *en* and 1:2; 1:1.5; 1:1; 1:0.75; 1:0.5; 1:0.25; 1:0.1; 1:0.05 for *deta*. The reaction mixtures were stirred for 24 h and after the reaction time, the amine-modified products were filtered, several times washed with methanol, dried, and further analyzed.

Methods and characterization. Infrared spectra (IR) of the prepared materials were measured on a Nicolet 6700 instrument in the wavelength range of 4000–400 cm^{–1} using the ATR method. Elemental analysis was measured on a CHNOS Elemental Analyzer Vario MICRO instrument with a sample weight of ~2 mg. Thermoanalytical experiments, thermogravimetry with simultaneous differential scanning calorimetry coupled with mass spectrometry (TG/DSC-MS) were performed using a SetsysEvolution (Setaram). The measurements were carried out with 16–19 mg of the sample using α-Al₂O₃ crucible. TG-DSC curves were recorded in the inert atmosphere of Ar (20 cm³ min^{–1}) from 15–800 °C with a heating rate of 10 °C min^{–1}. Thermal analysis with simultaneous differential thermal analysis (TG/DTA) was carried out on a Netzsch STA 449 F1 Jupiter. The sample with the weight of ~20 mg was placed in a corundum crucible and heated from 30 to 900 °C with a heating rate of 10 °C min^{–1} in the air atmosphere with a flow rate of 60 cm³ min^{–1}. Powder X-ray diffraction (PXRD) experiments were done in reflection geometry on a Rigaku MiniFlex 600 multipurpose diffractometer using Cu/Kα radiation (λ = 1.54056 Å) and 2θ continuous scan at 2° min^{–1} from 2 to 60°. Adsorption and desorption of Ar, CO₂ and H₂ were performed at – 186 °C, 0 °C, and – 196 °C, respectively, using the Autosorb iQ-XR (Quantachrome Instruments). Void volume was determined by a helium-free NOVA[®] approach³⁴. Samples were activated in a vacuum (20 h, 150 °C). The volume of micropores ($V_{p, \text{micro}}$, pore diameter < 2 nm) and mesopores ($V_{p, \text{meso}}$, pore diameter 2–50 nm) were calculated from pore size distribution curves obtained by fitting the Ar adsorption data by a Non-Local Density Functional Theory (NLDFT) adsorption kernel (ASiQwin software, Quantachrome Instruments) assuming cylindrical pores. BET area (S_{BET}) was calculated according to the procedure for microporous materials³⁵, based on which adsorption isotherm points in the region of ca. 10^{–3}–10^{–2} of p/p_0 (relative pressure) were used; 0.142 nm² as a cross-sectional area of Ar molecule was assumed.

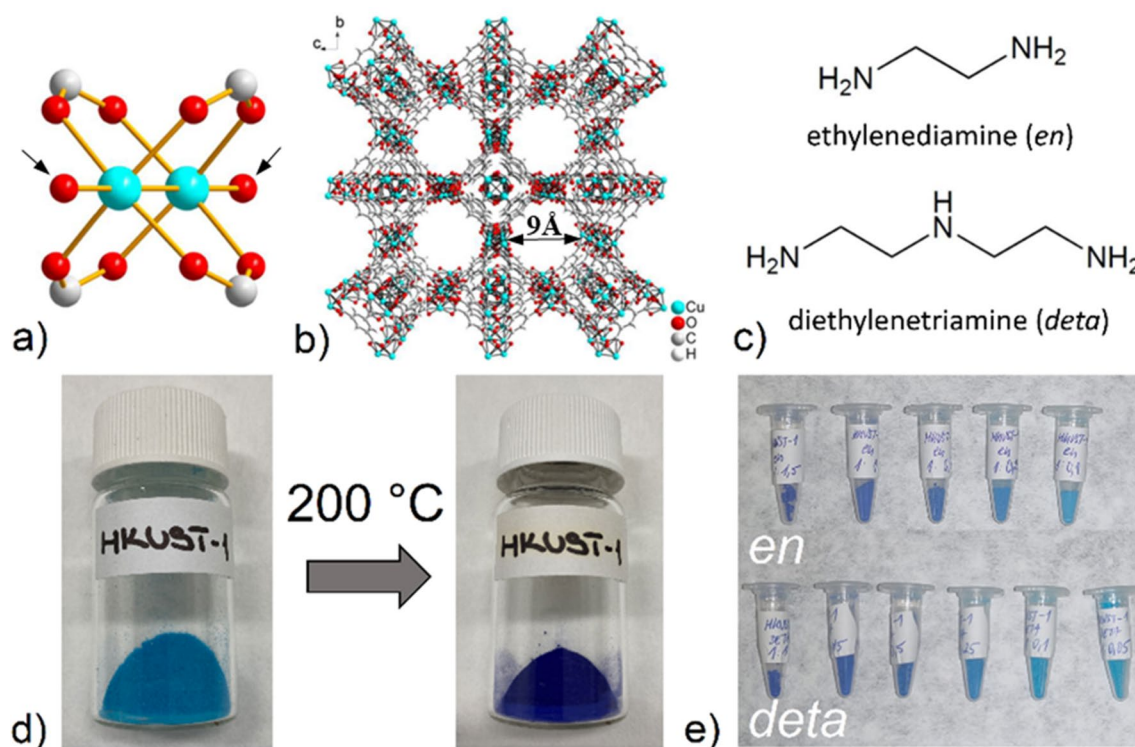


Figure 1. (a) Paddle-wheel cluster, where the arrows indicate the oxygen of the coordinated water molecules and (b) final framework of HKUST-1³⁶. (c) Molecular structure of amines (*en* and *deta*) used in post-synthetic modification. (d) Solvatochromism present in activated HKUST-1 at 200 °C. (e) A view of the prepared materials and their different colors depending on the molar ratio of amines used.

Results and discussion

HKUST-1 structure and synthesis. HKUST-1 is one of the first MOF compounds, which attract scientists around the world in the preparation and application of metal–organic structures³⁶. The crystal structure of HKUST-1 consists of two copper(II) cations with coordination number five, which form a paddle-wheel cluster with a square secondary building unit (SBU). Four carboxylate groups are coordinated to the cluster in *syn-syn* mode, forming a tetragonal base, and the fifth Cu(II) coordination position is occupied by coordinated water molecules to form a tetragonal pyramid-shaped polyhedron, as shown in Fig. 1a. The mutual combination of the mentioned building blocks results in the final 3D polymeric framework of HKUST-1 with bimodal pore size distribution, small cage with diameter 3.5 Å and large cage with diameter 9 Å (see Fig. 1b). The post-synthetic modification of HKUST-1 consisted of replacing coordinated water molecules in the paddle-cluster with amine molecules with different numbers of amine functional groups (ethylenediamine (*en*, diamine) and diethylenetriamine (*deta*, triamine)), see Fig. 1c. Water molecules were removed by heating the compound at 200 °C to form coordinatively unsaturated sites (CUSs) to which amines were attached via nitrogen donor atoms. Activation of the material was accompanied by a color change (from blue to purple, see Fig. 1d) due to solvatochromism, which is caused by a change in the donor set and coordination environment of the central atoms. The selected molar ratios for Cu(II) in HKUST-1: *en* were 1:2; 1:1.5; 1:1; 1:0.5; 1:0.25; 1:0.1 and for Cu(II) in HKUST-1: *deta* were 1:2; 1:1.5; 1:1; 1:0.75; 1:0.5; 1:0.25; 1:0.1; 1:0.05. At high molar ratios (= high amine concentrations), HKUST-1 was decomposed due to a strongly alkaline environment. In the case of *en*, the decomposition of HKUST-1 occurred at a molar ratio of 1:2 and *deta* at ratios of 1:2 and 1:1.5. At medium ratios (1:1.5; 1:1 for *en* and 1:1; 1:0.75; 1:0.5 for *deta*) there was no decomposition of HKUST-1, but its color change (from blue to violet) was observed. At low amine concentrations (1:0.1; 1:0.25; 1:0.5 for *en* and 1:0.05; 1:0.1; 1:0.25 for *deta*) the materials retained the original color as the original HKUST-1, as depicted in Fig. 1e.

Characterization. Infrared spectroscopy (IR), thermal analysis (TG / DSC) and powder X-ray diffraction (PXRD) methods were used to identify HKUST-1, monitor the post-synthetic modification process, thermal stability and characterization of prepared materials.

The IR spectra of the materials are shown in Fig. 2a, Fig. S1 in ESI and the assignment of the characteristic vibration absorption bands are summarized in Table 1 in ESI. The comparison of IR spectra of as-synthesized (AS), solvent exchanged (EX) and activated (AC) HKUST-1 is shown in Fig. S1 in ESI and confirm successful ethanol exchange process and material's activation based on the presence/absence of characteristic solvent absorption bands. In all infrared spectra (see Fig. 2a), a broad absorption band at 3400 cm⁻¹ is present, which can be assigned to the OH stretching vibration ($\nu(\text{OH})$) of the physisorbed water molecules. BTC linker in the materials is evident due to the presence of several characteristic vibrations: $\nu(\text{COO}^-)_{\text{as}}$ about 1550 cm⁻¹, $\nu(\text{COO}^-)_{\text{s}}$

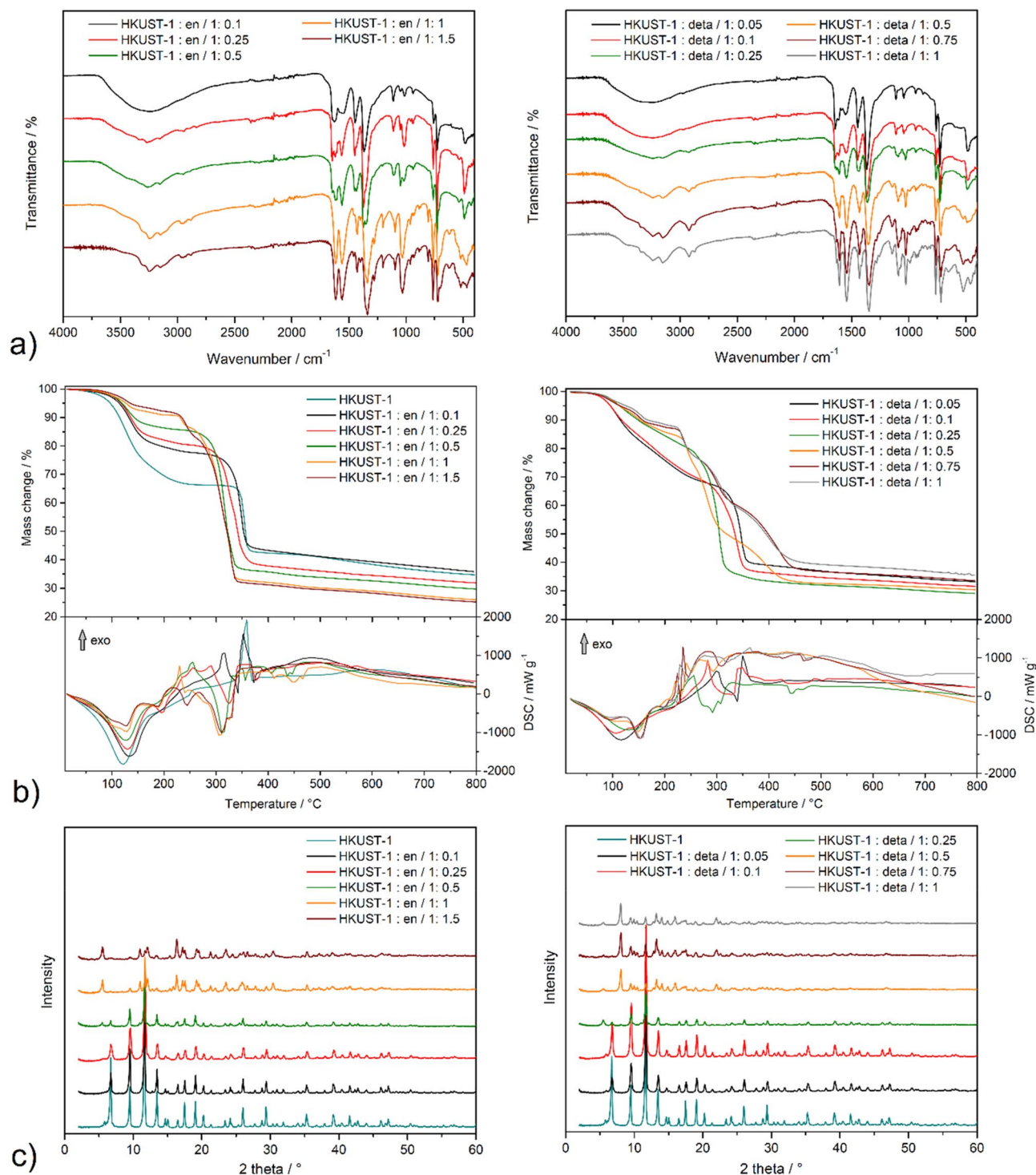


Figure 2. (a) Infrared spectra of HKUST-1 materials modified with *en* and *deta*. (b) TG/DSC curves of prepared materials measured in the temperature range of 20–800 $^{\circ}\text{C}$. (c) PXRD patterns of prepared materials in the 2θ range of 2–60 $^{\circ}$.

in the range of 1340–1370 cm^{-1} , aromatic $\delta(\text{CCH})_{\text{ar}}$ around 1100 cm^{-1} and $\delta(\text{COO}^-)$ about 720 cm^{-1} . For the modified materials, a gradual increase in intensity of $\nu(\text{NH})$ around 3200 cm^{-1} , $\delta(\text{NH})$ about 1580 cm^{-1} , several absorption bands of aliphatic $\nu(\text{CH})_{\text{aliph}}$ under 3000 cm^{-1} can be observed in the IR spectra with increasing concentration of amines, which are characteristic for *en* and *deta* (see Fig. 2a, Table 1 in ESI).

Thermal robustness and behavior of prepared materials were investigated by a combination of thermogravimetric (TG) analysis and differential scanning calorimetry (DSC) (see Fig. 2b). Unmodified HKUST-1 is thermally stable up to 50 $^{\circ}\text{C}$, above mentioned temperature the compound desolvates in the temperature range

Material	Ar @ - 186 °C			CO ₂ @ 0 °C			H ₂ @ - 196 °C		
	S _{BET} (m ² g ⁻¹)	V _{p, micro} (cm ³ g ⁻¹)	V _{p, meso} (cm ³ g ⁻¹)	V (cm ³ g ⁻¹)	w (wt.%)	n (mmol g ⁻¹)	V (cm ³ g ⁻¹)	W (wt.%)	n (mmol g ⁻¹)
HKUST-1	1468	0.681	0.000	77.3	15.19	3.45	165.3	1.49	7.37
HKUST-1: <i>en</i> /1:0.1	656	0.321	0.063	113.6	22.31	5.07	202.1	1.82	9.02
HKUST-1: <i>en</i> /1:0.25	511	0.251	0.067	88.8	17.45	3.96	155.7	1.40	6.95
HKUST-1: <i>en</i> /1:0.5	309	0.147	0.068	46.7	9.18	2.09	82.9	0.75	3.70
HKUST-1: <i>en</i> /1:1	53	0.025	0.007	11.4	2.24	0.51	11.3	0.10	0.50
HKUST-1: <i>en</i> /1:1.5	–	–	–	3.1	0.61	0.14	6.5	0.06	0.29
HKUST-1: <i>deta</i> /1:0.05	1439	0.671	0.034	168.4	33.09	7.52	253.8	2.28	11.33
HKUST-1: <i>deta</i> /1:0.1	1245	0.577	0.050	156.7	30.79	7.00	236.4	2.12	10.55
HKUST-1: <i>deta</i> /1:0.25	769	0.377	0.034	93.5	18.37	4.17	156.9	1.41	7.00
HKUST-1: <i>deta</i> /1:0.5	311	0.146	0.011	37.7	7.41	1.68	67.9	0.61	3.03
HKUST-1: <i>deta</i> /1:0.75	90	0.043	0.010	14.0	2.74	0.62	44.0	0.40	1.96
HKUST-1: <i>deta</i> /1:1	7.2	–	–	5.4	1.06	0.24	36.9	0.33	1.65

Table 1. Obtained and calculated results of textural properties determined from Ar adsorption measurements @ - 186 °C and storage capacity of CO₂ and H₂ measured 0 °C and - 196 °C, respectively of prepared materials in different units. S_{BET} BET area, V_p pore volume, V adsorbed gas volume, w mass percent, n amount of substance, - undetectable, micro micropore, meso mesopore.

50–250 °C with a weight loss of 33.5 wt.% on TG. The activated form is stable up to 330 °C, as it seems by the plateau on TG curve, and subsequently, in the interval 330–380 °C thermal decomposition of the framework occurs with a mass change of 23.2 wt.%. For *en*-modified materials in a molar ratio of 1:0.1 to 1:0.5, the shape and course of the TG curves are identical to unmodified HKUST-1, and the thermal decomposition takes place in two steps (see Fig. 2b, left). From the TG curves of mentioned materials, it is evident that they contain less amount of solvents in the cavities (15.2–22.6 wt.%, range 20–265 °C), and higher weight loss is observed in the thermolysis of the frameworks (33.6–48.5 wt.%, range 265–385 °C) compared to HKUST-1. This observation can be explained by the increasing content of *en* molecules in materials, which reduce the free void volume for solvents and increase the content of organic building blocks in the compounds. Several changes on the TG curves of HKUST-1: *en*/1:1 and HKUST-1: *en*/1:1.5 are observed: smaller amounts of solvents (8.3–9.4 wt.%, range 20–220 °C), thermal decomposition takes place in three steps, higher content of organic components (57.9–59.8 wt.%) and reduced thermal stability of the organic part (220–350 °C). Regarding the thermal stability of *en*-modified materials, it should be noted that they show lower thermal stability of the framework compared to HKUST-1. Similar thermal behavior was observed for the *deta*-modified materials as for the *en*-functionalized samples (see Fig. 2b, right). The course of TG curves HKUST-1: *deta* with molar ratios of 1:0.05, 1:0.1 and 1:0.25 show a similar trend as unmodified HKUST-1 (desolvation 20.5–30.6 wt.%, range 20–250 °C and organic part 29.9–45.6 wt.%, range 250–370 °C). Increasing the concentration of *deta* is associated with the course changes of the TG curves on which five decomposition steps are observed and indicate a significant chemical change of the prepared materials. In general for all thermogravimetric curves, in the first thermal decomposition steps, desolvation/dehydration occurred in the mass change interval of 8.1–34.2 wt.% on TG curves, which are accompanied by endothermic effects on DSC curves with peaks in the range of 96–154 °C. Pyrolysis of the organic moiety occurred in the 220–450 °C temperature range, which is followed by exothermic effects on the DSC in the range of 220–370 °C. According to the changes in TG curves and colour change of materials with higher amine content, we assumed a phase change of these compounds, which was confirmed by PXRD measurements (see text below).

The structural and thermal stability of HKUST-1 and materials after functionalization with *en* and *deta* in different molar ratios was monitored by PXRD measurements (see Fig. 2c, Figs. S2–S5 in ESI). In Fig. S2 in ESI the comparison of measured and calculated PXRD (Chui et al. 1999) patterns of HKUST-1 is presented, which are in good agreement and confirm the successful preparation of HKUST-1. Heating PXRD measurements studied the thermal stability of HKUST-1 at 50, 100, 150, 200, 250 and 350 °C (see Fig. S3 in ESI), which showed that HKUST-1 is stable after the activation process and its decomposition occurs at 350 °C. The obtained results are in good agreement with TG measurements (see Fig. 2b). From the PXRD patterns of HKUST-1: *en* materials (see Fig. 2c, left), it is evident that at molar ratios of 1:0.1 and 1:0.25 only the HKUST-1 phase is present. At a ratio of 1:0.5, in addition to the diffraction peaks corresponding to HKUST-1, novel diffraction lines appear, corresponding to the formation of a new phase. A further increase in the concentration of *en* completely eliminates HKUST-1, while only the diffraction peaks of the novel phase are present in the PXRD patterns. Similar changes were observed for *deta*-modified materials (see Fig. 2c, right). The HKUST-1 phase is stable up to a molar ratio

of 1:0.1, at a ratio of 1:0.25, a mixture consisting of HKUST-1 and a new phase is present, and at higher molar ratios, only a novel phase is present. The described phase changes were manifested visually by a colour change of prepared materials from blue to purple (see Fig. 1e). The thermal stability of selected *en*-modified materials at a low ratio, HKUST-1: *en*/1:0.25 represented by HKUST-1 phase (see Fig. S4 in ESI) and a high ratio, HKUST-1: *en*/1:1 characterised by novel phase (see Fig. S5 in ESI) was also studied. For the low ratio material, the same thermal stability as pure HKUST-1 was observed, and the high molar sample is thermally stable up to 100 °C, at higher temperatures, a phase change occurs, and at 350 °C the material decomposed.

The new phase that forms at high amine concentrations (specifically HKUST-1: *en*/1:1.5) was studied through various physicochemical techniques in order to identify it and determine its chemical composition. By searching the CCDC (Cambridge Crystallographic Data Center) database, it can be concluded that two compounds containing Cu(II), BTC(-III) ions and a derivative of *en* ligand with the chemical composition $\text{Cu}_4(\text{HBTC})_4(\text{tmen})_4 \cdot 12\text{H}_2\text{O}$ (FUHYAJ³⁷) and $\text{Cu}_3(\text{BTC})_2(\text{tmen})_3(\text{H}_2\text{O})_2 \cdot 6.5\text{H}_2\text{O}$ (FEXSOR³⁸) were synthesized (see Fig. 3a). In their preparation, a *N*-tetramethyl-substituted *en* ligand (*tmen*—*N,N,N',N'*-tetramethylethylenediamine) on nitrogen atoms was used. By comparing the PXRD pattern of HKUST-1: *en*/1:1.5 material and the mentioned compounds, a similarity with the compound $\text{Cu}_3(\text{BTC})_2(\text{tmen})_3(\text{H}_2\text{O})_2 \cdot 6.5\text{H}_2\text{O}$ can be observed (see Fig. 3b). Since it was not possible to clearly identify the newly formed phase by the PXRD comparison, we decided to study the exact chemical composition of the modified material HKUST-1: *en*/1:1.5. The presence of organic building blocks (BTC and *en*) was confirmed by IR spectroscopy. The IR spectrum (Fig. 3c) of compound contains characteristic absorption bands for the carboxylate group of the BTC linker ($\nu(\text{COO}^-)_{\text{as}}$ 1563 cm^{-1} , $\nu(\text{COO}^-)_{\text{as}}$ 1337 cm^{-1} and $\delta(\text{COO}^-)$ 723 cm^{-1}) and the presence of *en* is confirmed by typical bands for aliphatic CH_2 ($\nu(\text{CH})_{\text{al}}$ 2970, 2947, 2927 and 2884 cm^{-1}) and NH_2 groups ($\nu(\text{NH})$ 3248 and 3148 cm^{-1} ; $\delta(\text{NH})$ 1615 cm^{-1}). By TG-MS, the thermal stability of the new phase was studied in an inert atmosphere of argon, through which the presence of *en* was also confirmed (see Fig. 3d). In the temperature interval 60–220 °C, the release of water molecules (dehydration) occurs, which was manifested on the MS spectrum by the signal $m/z = 18$. In the second step of thermal decomposition in the temperature interval 220–280 °C, a mass loss is observed, which corresponds to the release of coordinated *en* molecules with a signal of $m/z = 60$. Further heating in the temperature range of 280–350 °C results in pyrolysis of the polymer skeleton and further release of *en*. The mass spectra of gaseous products in the mentioned interval showed the signals with $m/z = 18, 32, 44, 60$ corresponding to $\text{H}_2\text{O}, \text{CO}, \text{CO}_2$ and *en*, respectively. The above results show that the thermal decomposition of *en* and BTC cannot be clearly distinguished, and both decomposition steps take place simultaneously. As the thermal analysis was carried out in an inert atmosphere of argon, the sample did not fully decompose, and carbonized material was present in the resulting thermal decomposition product. For this reason, TG/DTA measurements were carried out in an oxidizing air atmosphere to ensure complete thermolysis of the material, and based on weight losses and CHN elemental analysis results, it was possible to determine the exact chemical composition of the sample HKUST-1: *en*/1:1.5. TG/DTA curves of HKUST-1: *en*/1:1.5 measured in the air atmosphere are shown in Fig. 3e. In the first thermal decomposition step, in the temperature interval 40–150 °C, a mass loss of 10.18% was observed, which corresponds to the release of five water molecules (clcd. mass change 10.29%). Dehydration of the sample was manifested on the DTA curve by two endothermic effects at 96 and 118 °C. In the following two overlapping decomposition steps, the release of three amine molecules and the thermal decomposition of two BTC molecules occur. The total mass loss corresponding to the organic components was 61.77% (clcd. mass change 62.45%), and their decomposition/release was accompanied by three exothermic effects on the DTA curve at 244, 324 and 377 °C. PXRD identified the resulting thermal decomposition product as CuO (see PXRD pattern in Fig. 3f) with the experimental residual mass of 27.94% (clcd. residual mass 27.27%). Based on the described weight losses, it was possible to determine the exact chemical composition of the material HKUST-1: *en*/1:1.5 with the formula $\text{Cu}_3(\text{BTC})_2(\text{en})_3 \cdot 5\text{H}_2\text{O}$. The chemical composition was also confirmed by CHN elemental analysis, while the calculated and measured percentages of chemical elements in the compound are in good agreement ($M_r = 875.25 \text{ g mol}^{-1}$; exp.: C 32.54%, H 4.51%, N 21.92%, clcd.: C 32.93%, H 4.61%, N 21.78%).

Texture and gas adsorption properties. After successful preparation and characterization of the amine-modified compounds, the materials' textural properties (S_{BET} area, micropore and mesopores volume) were studied by argon adsorption @ – 186 °C, and their storage capacities of carbon dioxide @ 0 °C and hydrogen @ – 196 °C were investigated. The results obtained from the adsorption measurements are shown in Fig. 4 and summarized in Table 1.

Argon adsorption measurements @ – 186 °C studied the effect of post-synthetic amine-modification on the textural properties of HKUST-1. Figure 4a shows a comparison of the Ar adsorption/desorption isotherms of pristine HKUST-1, *en*- and *deta*-modified materials, and the selected textural properties calculated from the respective isotherms are summarized in Table 1. The materials show a typical type Ia isotherm classified by IUPAC³⁵, characteristic for materials with narrow micropores below ca. 1 nm in diameter. This is in line with obtained micropore sizes that were the same for all studied materials and ranged from 0.6 to 1.2 nm. As can be seen from the obtained results, the S_{BET} area of pristine HKUST-1 was 1468 $\text{m}^2 \text{ g}^{-1}$, and after modification, the area of materials decreases with increasing amine concentration. The S_{BET} area of modified materials decreased from HKUST-1: *en*/1:0.1 (656 $\text{m}^2 \text{ g}^{-1}$) to undetectable values for HKUST-1: *en*/1:1.5 for *en*-modified samples, and in *deta*-functionalized materials from HKUST-1: *deta*/1:0.05 (1439 $\text{m}^2 \text{ g}^{-1}$) to undetectable values for HKUST-1: *deta*/1:1. The observed phenomenon can be explained by the reduction in the free pore volume/area of the modified materials due to the binding of a larger number of molecules and the bulk of amine molecules themselves. It can be seen that even a low molar ratio of *en* (1:0.1) in HKUST-1 structure leads to a halving of textural properties like S_{BET} . On the contrary, the same content of *deta* leads to only a small decrease in texture

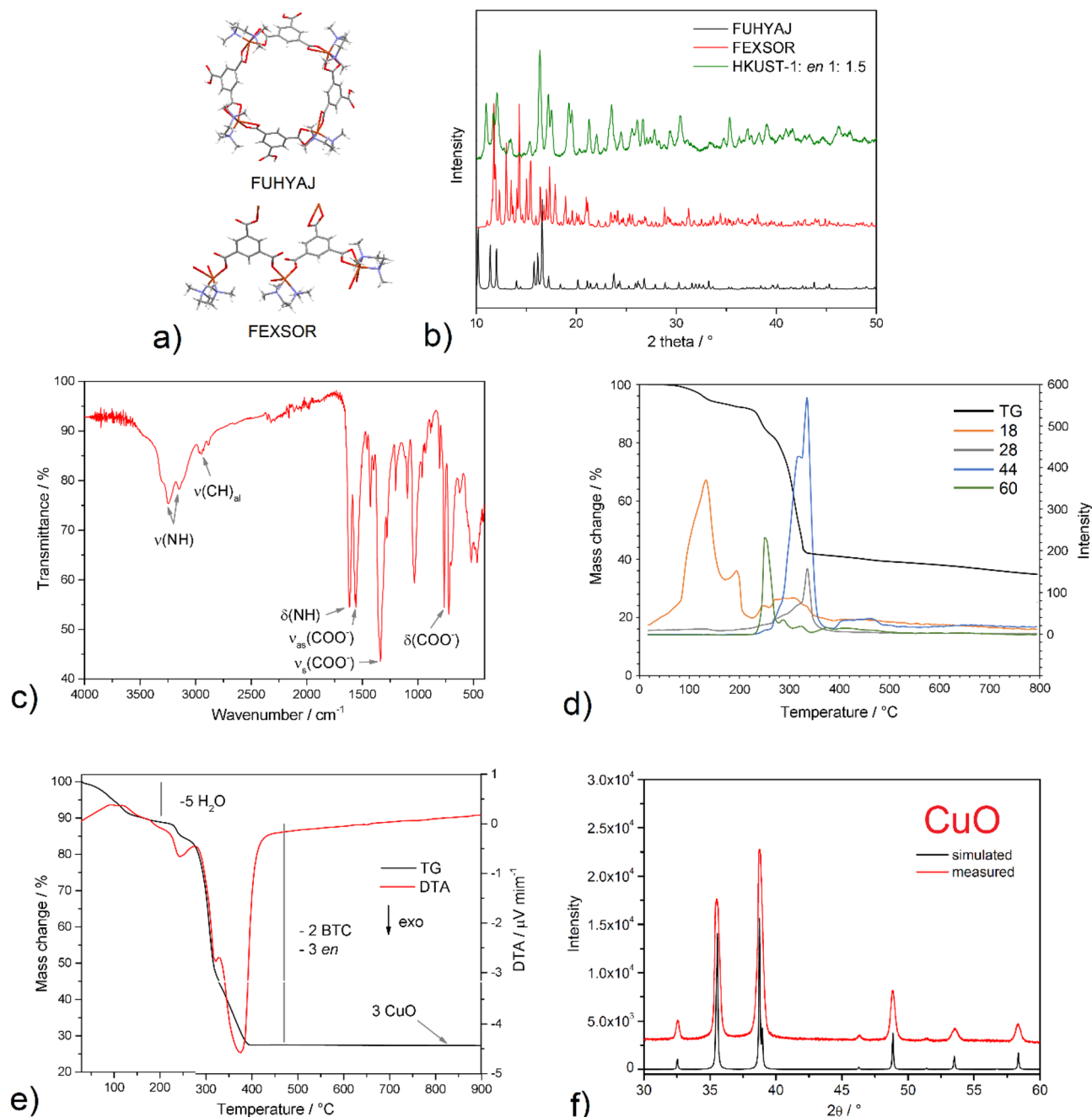


Figure 3. (a) The crystal structure of $\text{Cu}_4(\text{HBTC})_4(\text{tmen})_4 \cdot 12\text{H}_2\text{O}$ (FUHYAJ) and $\text{Cu}_3(\text{BTC})_2(\text{tmen})_3(\text{H}_2\text{O})_2 \cdot 6.5\text{H}_2\text{O}$ (FEXSOR). (b) Comparison of measured PXRD pattern of sample HKUST-1: *en*/1:1.5 and calculated patterns of $\text{Cu}_4(\text{HBTC})_4(\text{tmen})_4 \cdot 12\text{H}_2\text{O}$ (FUHYAJ) and $\text{Cu}_3(\text{BTC})_2(\text{tmen})_3(\text{H}_2\text{O})_2 \cdot 6.5\text{H}_2\text{O}$ (FEXSOR). The detailed characterisation of HKUST-1: *en*/1:1.5 using (c) infrared spectroscopy, (d) TG-MS in an argon atmosphere, (e) TG/DTA in an air atmosphere, (f) Identification of final thermal decomposition product by PXRD.

parameters ($1245 \text{ m}^2 \text{ g}^{-1}$). The observed difference can be explained based on the length of amines used and their coordination modes to coordinatively unsaturated sites (CUSs) within the HKUST-1 framework. The size of *en* molecule is 3.7 \AA , and *deta* molecule is 6.3 \AA (see Fig. S6a in ESI). Since the smallest distance between two Cu(II) ions within the HKUST-1 skeleton is 5.2 \AA (see Fig. S6b in ESI), *deta* molecules can bridge and connect the two metal centres, while *en* can only bind terminally. The mentioned coordination modes of amine molecules are schematically drawn in Fig. S6c in ESI. Fig. S6c in ESI shows the entrance window to the HKUST-1 porous framework, where four Cu(II) cations are located inside the pore. When amines are coordinated to the metal centre, four molecules of *en* or two molecules of *deta* are coordinated within the entrance window, as shown in Fig. S6c in ESI. Thus, in the case of *en* coordination, the free pore volume is more efficiently filled, resulting in a lower surface area than bulkier *deta* molecules. To confirm the assumed theory, we prepared HKUST-1 modified

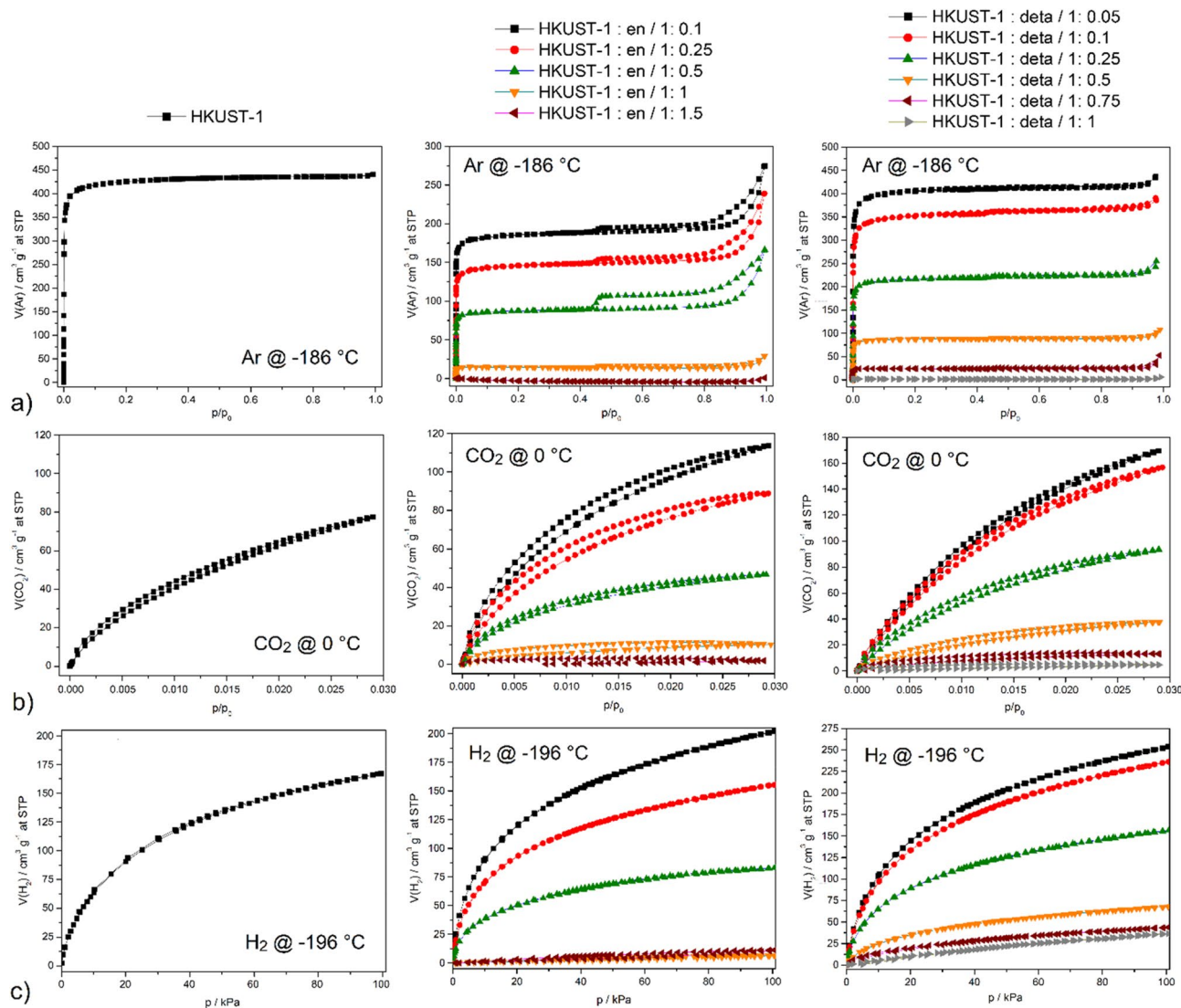


Figure 4. Adsorption/desorption isotherms of (a) argon @ $-186\text{ }^{\circ}\text{C}$, (b) carbon dioxide @ $0\text{ }^{\circ}\text{C}$ and (c) hydrogen @ $-196\text{ }^{\circ}\text{C}$ on pristine HKUST-1, *en* and *deta*-modified HKUST-1 materials.

material with *bapen* (*bapen* = *N,N'*-bis(3-aminopropyl)-1,2-diaminoethane). *Bapen* is a dimensionally longer ligand compared to *en* and *deta* and contains four amine groups connected by ethylene and propylene bridges (see inset in Fig. S7 in ESI). By comparing the surface area of modified HKUST-1 materials: HKUST-1: *en*/1:0.1 ($656\text{ m}^2\text{ g}^{-1}$) < HKUST-1: *bapen*/1:0.1 ($1028\text{ m}^2\text{ g}^{-1}$) < HKUST-1: *deta*/1:0.1 ($1245\text{ m}^2\text{ g}^{-1}$), it can be concluded that in material HKUST-1: *bapen*/1:0.1 extended amine ligand also did not cause a significant decrease in S_{BET} as in the case of *en* (see Fig. S7 in ESI).

When reducing the textural properties of materials due to the increase in the concentration of amines, it is also necessary to consider the results of PXRD analysis. PXRD measurements revealed the formation of a new phase, which is present in the mixture at medium molar ratios and in pure form at the highest ratios. As is evident from the measured and calculated textural properties, the newly formed phase is non-porous, as the materials HKUST-1: *en*/1:1, HKUST-1: *en*/1:1.5, HKUST-1: *deta*/1:0.75 and HKUST-1: *deta*/1:0.75 adsorb only minimally amounts of argon. Moreover, on argon adsorption isotherms an increase in the Ar adsorbed volume at higher relative pressures is observed, as well as H4 hysteresis loop typical for aggregated crystals³⁵ indicating the presence of mesopores especially in the *en*-modified materials with ratios from 1:0.1 to 1:0.5 containing low concentration of *en*. Presence of mesopores or larger pores can be explained by formation of intraggregate voids³⁹ due to surface modification, or etching of the material caused by increasing the basicity of the environment during the modification process. As shown in Table 1, the percentage increase in mesopore volume ($V_{p,\text{micro}}/V_{p,\text{meso}}$) rises with increasing amine concentration, and the values increase from 20 to 46% for HKUST-1: *en* and from 5 to 23% for HKUST-1: *deta*. Similar changes in textural properties were observed for other amine-modified MOF materials such as MOF-74²³, MOF-177³⁹ and IRMOF-74⁴⁰.

From the measured carbon dioxide adsorption/desorption isotherms, an increasing amount of stored CO_2 can be generally observed with the decreasing concentration of amines (see Fig. 4b, Table 1). The pristine

HKUST-1 material can store in its framework 15.19 wt.% (3.45 mmol g⁻¹) of CO₂ @ 0 °C and 1 bar. The maximum CO₂ storage capacity of *en*-modified materials was measured on HKUST-1: *en*/1:0.1 with a value of 22.31 wt.% (5.07 mmol g⁻¹) and decreased to 0.61 wt.% (0.14 mmol g⁻¹) for HKUST-1: *en*/1:1.5 @ 0 °C and 1 bar. For *deta*-modified materials, the highest CO₂ uptake of 33.09 wt.% (7.52 mmol g⁻¹) was obtained for HKUST-1 : *deta*/1:0.05 and the lowest value of 1.06 wt.% (0.24 mmol g⁻¹) for HKUST-1 : *deta*/1:1 @ 0 °C and 1 bar. In the case of CO₂ adsorption, it is necessary to consider the competitive processes of chemisorption carried out on amine groups (R-NH₂+CO₂=R-NH₂COO) and physisorption, which takes place in the void pore volume of materials. As already discussed in the context of Fig. S6, the void volume seems to be larger for materials modified by *deta*, than *en*, which is in line with the fact that *deta*-modified material (e.g. 1:0.1) exhibits larger CO₂ adsorption capacity compared to material modified by an equivalent quantity of *en*. In terms of concentration, although the number of active sites for the sorption of CO₂ molecules increases with the increasing concentration of coordinated amines, the free pore volume within the framework decreases due to the bulkiness of amines. The phenomenon described is clear from the measured results of CO₂ adsorption for amine-modified materials (see Fig. 4b, Table 1). Another fact that needs to be considered is the already mentioned phase change of materials at higher amine molar ratios, leading to non-porous materials' formation. A trend of decreasing CO₂ adsorption capacity with increasing amine concentration was also observed for MOF-177 functionalized with polyethyleneimine (PEI) and *deta*³⁹. CO₂ storage capacity decreased for PEI modified materials in following order: MOF-177-PEI (10%) (12.8 wt.%) > MOF-177-PEI (20%) (10.6 wt.%) > MOF-177-PEI (30%) (9.7 wt.%) and *deta*-modified materials in the order: MOF-177-*deta* (20%) (12.3 wt.%) > MOF-177-*deta* (30%) (9.6 wt.%) @ 25 °C and 1 bar. The UiO-66 material modified with PEI could also be mentioned, whose CO₂ capacities decreased in the order of 11.9 wt.% for 20PEI@UiO-66 to 9.2 wt.% for 40PEI@UiO-66 @ 25 °C and 1 bar⁴¹. It can be summarized that the most promising material of our study with the highest adsorption capacity is HKUST-1: *deta*/1:0.05, which is able to adsorb 33.09 wt.% of CO₂ (168.4 cm³ g⁻¹, 7.52 mmol g⁻¹).

Because hydrogen is a nonpolar and the smallest known molecule, the storage of large amounts of H₂ is difficult, and for this reason, various MOF materials are intensively studied. It is known that efficient hydrogen storage can be achieved by the presence of alkali and alkaline earth metals within the framework, by the insertion of metal nanoparticles, or by post-synthetic modification^{5,25,26}. Efficient hydrogen storage can also occur due to steric effects, as the calculated ideal pore size for hydrogen storage is 6–7 Å^{32,33}. For this reason, the effect of different molar ratios of different bulk amines on the pore size in HKUST-1 was investigated to store H₂ efficiently. The measured hydrogen adsorption isotherms of the prepared materials @ – 196 °C are shown in Fig. 4c, and the achieved H₂ storage capacities in different units are listed in Table 1. As with Ar and CO₂ adsorption, the same trend and thus increasing in adsorbed amounts of H₂ with decreasing concentration of amines. The highest hydrogen uptakes were measured for HKUST-1 : *en*/1:0.1 and HKUST-1: *deta*/1:0.05 with 1.82 and 2.28 wt.%, respectively @ – 196 °C and 1 bar. The pore size can explain the increased affinity of H₂ for the prepared materials. Calculated pore size distributions (PSD) were obtained by fitting the Ar adsorption data using a NLDFT adsorption kernel assuming cylindrical pores. All materials contain pore sizes ranging from 0.6 to 1.2 nm, covering the ideal pore size range of 0.6–0.7 nm. The decreasing H₂ storage capacity of materials with increasing concentration of amines is due to the decreasing textural parameters like S_{BET} and $V_{p, micro}$. The formation of a non-porous phase cannot be neglected in materials with a higher concentration of amines, through which it is possible to explain the decreasing H₂ storage capacity of the compounds. According to the best of our knowledge, the effect of stored hydrogen on amine-modified MOFs has not yet been studied and the present study thus represents a pilot project. There are published articles dealing with the separation of H₂/CO₂ on amine-modified Mg-MOF-74 or MIL-53^{42,43}, but not for H₂ storage. In conclusion, however, it can be concluded that the measured highest storage capacities are in the range for top MOF materials^{25,26} and are comparable, for example, with UiO-66(Zr)⁴⁴, MOF-519⁴⁵, Mg-MOF-74⁴⁶ with maximal H₂ uptakes of 1.87 wt.%, 2.13 wt.% and 2.20 wt.%, respectively @ – 196 °C and 1 bar.

Conclusion

The intention of present study was to increase the adsorption capacity of carbon dioxide and hydrogen in HKUST-1 through post-synthetic modification of amines with different numbers of amine groups. Ethylenediamine (*en*, diamine) and diethylenetriamine (*deta*, triamine) were chosen as amines in this study. At present, CO₂ capture is realized using liquid amines, which can efficiently bind CO₂ to form carbamates, but have many disadvantages (corrosivity, regeneration temperature). For this reason, a strategy of anchoring amines on porous support represented by the well-known MOF material, HKUST-1, was chosen. From the HKUST-1 stability point of view, it can be summarized that with increasing molar ratios of HKUST-1: amine, decomposition (1:2 for *en* and 1:2; 1:1.5 for *deta*), or a phase change (1:1.5; 1:1; 1:0.5 for *en* and 1:1; 1:0.75; 1:0.5; 1:0.25 for *deta*) were observed, which led to the formation of a non-porous materials, that have been thoroughly investigated. From the experimentally obtained adsorption measurements, it can be concluded that with the increasing concentration of amines, the CO₂ capacity of materials decreased and that the *deta*-modified materials showed higher stored amounts of CO₂ (probably due to their higher void volume) compared to *en*-modified compounds. The highest storage capacities were obtained for HKUST-1 : *en*/1:0.1 and HKUST-1 : *deta*/1:0.05 materials with 22.31 wt.% and 33.09 wt.% of CO₂ @ 0 °C and 1 bar, respectively. The same trend was observed in hydrogen adsorption measurements, H₂ storage capacity for HKUST-1: *en* 1:0.1 was 1.82 wt.% @ – 196 °C and 1 bar and for HKUST-1: *deta* 1:0.05 was 2.28 wt.% @ – 196 °C and 1 bar are comparable to other top MOF materials.

Data availability

The data are available on request, please contact corresponding author via e-mail (miroslav.almasi@upjs.sk).

Received: 9 August 2022; Accepted: 12 October 2022

Published online: 17 October 2022

References

- Link 1, https://ec.europa.eu/info/strategy/priorities-2019-2024/european-green-deal/delivering-european-green-deal_en.
- Link 2, <https://www.osti.gov/biblio/1480987-cost-performance-baseline-fossil-energy-plants-volume-bituminous-coal-pc-natural-gas-electricity-revision>.
- Link 3, <https://www.energy.gov/eere/fuelcells/doe-technical-targets-onboard-hydrogen-storage-light-duty-vehicles>.
- Fan, W., Zhang, X., Kang, Z., Liu, X. & Sun, D. Isoreticular chemistry within metal-organic frameworks for gas storage and separation. *Coord. Chem. Rev.* **443**, 213968. <https://doi.org/10.1016/j.ccr.2021.213968> (2021).
- Almási, M., Zelenák, V., Gyepes, R., Zauska, U. & Bourrelly, S. A series of four novel alkaline earth metal-organic frameworks constructed of Ca(II), Sr(II), Ba(II) ions and tetrahedral MTB linker: Structural diversity, stability study and low/high-pressure gas adsorption properties. *RSC Adv.* **10**, 32323–32334. <https://doi.org/10.1039/d0ra05145d> (2020).
- Fakhraei Ghazvini, M., Vahedi, M., Najafi Nobar, S. & Sabouri, F. Investigation of the MOF adsorbents and the gas adsorptive separation mechanisms. *J. Environ. Chem. Eng.* **9**, 104790. <https://doi.org/10.1016/j.jece.2020.104790> (2021).
- Almási, M., Király, N., Zelenák, V., Vilková, M. & Bourrelly, S. Zinc(II) and cadmium(II) amorphous metal-organic frameworks (aMOFs): Study of activation process and high-pressure adsorption of greenhouse gases. *RSC Adv.* **11**, 20137–20150. <https://doi.org/10.1039/d1ra02938j> (2021).
- Hao, M., Qiu, M., Yang, H., Hu, B. & Wang, X. Recent advances on preparation and environmental applications of MOF-derived carbons in catalysis. *Sci. Total Environ.* **760**, 143333. <https://doi.org/10.1016/j.scitotenv.2020.143333> (2021).
- Almási, M., Zelenák, V., Opanasenko, M. V. & Čejka, J. Efficient and reusable Pb(II) metal-organic framework for Knoevenagel condensation. *Catal. Lett.* **148**, 2263–2273. <https://doi.org/10.1007/s10562-018-2471-8> (2018).
- Yao, M. S., Li, W. H. & Xu, G. Metal-organic frameworks and their derivatives for electrically-transduced gas sensors. *Coord. Chem. Rev.* **426**, 213479. <https://doi.org/10.1016/j.ccr.2020.213479> (2021).
- Garg, A. *et al.* Metal-organic framework MOF-76(Nd): Synthesis, characterization and study of hydrogen storage and humidity sensing. *Front. Energy Res.* **8**, 604735. <https://doi.org/10.3389/fenrg.2020.604735> (2021).
- Garg, A. *et al.* Gd(III) metal-organic framework as an effective humidity sensor and its hydrogen adsorption properties. *Chemosphere* **305**, 135467. <https://doi.org/10.1016/j.chemosphere.2022.135467> (2022).
- Zhong, X. *et al.* Gd(III)-based inorganic polymers, metal-organic frameworks and coordination polymers for magnetic refrigeration. *CrystEngComm* **24**, 2370–2382. <https://doi.org/10.1039/d1ce01633d> (2022).
- Zelenák, V. *et al.* Large and tunable magnetocaloric effect in gadolinium-organic framework: Tuning by solvent exchange. *Sci. Rep.* **9**, 15572. <https://doi.org/10.1038/s41598-019-51590-2> (2019).
- Mallakpour, S., Nikkhoo, E. & Hussain, C. M. Application of MOF materials as drug delivery systems for cancer therapy and dermal treatment. *Coord. Chem. Rev.* **451**, 214262. <https://doi.org/10.1016/j.ccr.2021.214262> (2022).
- Almási, M., Zelenák, V., Palotai, P., Beňová, E. & Zelenáková, A. Metal-organic framework MIL-101(Fe)-NH₂ functionalized with different long-chain polyamines as drug delivery system. *Inorg. Chem. Commun.* **93**, 115–120. <https://doi.org/10.1016/j.inoche.2018.05.007> (2018).
- Rabiee, N. *et al.* Bioactive hybrid metal-organic framework (MOF)-based nanosensors for optical detection of recombinant SARS-CoV-2 spike antigen. *Sci. Total Environ.* **825**, 153902. <https://doi.org/10.1016/j.scitotenv.2022.153902> (2022).
- Almási, M. A review on state of art and perspectives of metal-organic frameworks (MOFs) in the fight against coronavirus SARS-CoV-2. *J. Coord. Chem.* **74**, 2111–2127. <https://doi.org/10.1080/00958972.2021.1965130> (2021).
- Li, L., Yang, J., Li, J., Chen, Y. & Li, J. Separation of CO₂/CH₄ and CH₄/N₂ mixtures by M/DOBDC: A detailed dynamic comparison with MIL-100(Cr) and activated carbon. *Microporous Mesoporous Mater.* **198**, 236–246. <https://doi.org/10.1016/j.micromeso.2014.07.041> (2014).
- Mason, J. A. *et al.* Application of a high-throughput analyzer in evaluating solid adsorbents for post-combustion carbon capture via multicomponent adsorption of CO₂, N₂ and H₂O. *J. Am. Chem. Soc.* **137**, 4787–4803. <https://doi.org/10.1021/jacs.5b00838> (2015).
- Luo, J., Wang, J., Li, G., Huo, Q. & Liu, Y. Assembly of a unique octa-nuclear copper cluster-based metal-organic framework with highly selective CO₂ adsorption over N₂ and CH₄. *Chem. Commun.* **49**, 11433. <https://doi.org/10.1039/c3cc47462c> (2013).
- Ghanbari, T., Abnisa, F. & Wan Daud, W. M. A. A review on production of metal organic frameworks (MOF) for CO₂ adsorption. *Sci. Total Environ.* **707**, 35090. <https://doi.org/10.1016/j.scitotenv.2019.135090> (2020).
- Su, X. *et al.* Postsynthetic functionalization of Mg-MOF-74 with tetraethylenepentamine: Structural characterization and enhanced CO₂ adsorption. *ACS Appl. Mater. Interfaces* **9**, 11299–11306. <https://doi.org/10.1021/acsami.7b02471> (2017).
- Lin, Y., Yan, Q., Kong, C. & Chen, L. Polyethyleneimine incorporated metal-organic frameworks adsorbent for highly selective CO₂ capture. *Sci. Rep.* **3**, 1859. <https://doi.org/10.1038/srep01859> (2013).
- Shet, S. P., Shanmuga Priya, S., Sudhakar, K. & Tahir, M. A review on current trends in potential use of metal-organic framework for hydrogen storage. *Int. J. Hydrog. Energy* **46**, 11782–11803. <https://doi.org/10.1016/j.ijhydene.2021.01.020> (2021).
- Langmi, H. W., Ren, J., North, B., Mathe, M. & Bessarabov, D. Hydrogen storage in metal-organic frameworks: A review. *Electrochim. Acta* **128**, 368–392. <https://doi.org/10.1016/j.electacta.2013.10.190> (2014).
- Grünker, R. *et al.* A new metal-organic framework with ultra-high surface area. *Chem. Commun.* **50**, 3450. <https://doi.org/10.1039/c4cc00113c> (2014).
- Gómez-Gualdrón, D. A. *et al.* Evaluating topologically diverse metal-organic frameworks for cryo-adsorbed hydrogen storage. *Energy Environ. Sci.* **9**, 3279–3289. <https://doi.org/10.1039/c6ee02104b> (2016).
- Park, H. J., Lim, D., Yang, W. S., Oh, T. & Suh, M. P. A highly porous metal-organic framework: Structural transformations of a guest-free MOF depending on activation method and temperature. *Chem. Eur. J.* **17**, 7251–7260. <https://doi.org/10.1002/chem.201003376> (2011).
- Almási, M. *et al.* Cytotoxicity study and influence of SBA-15 surface polarity and pH on adsorption and release properties of anticancer agent pemetrexed. *Mater. Sci. Eng. C* **109**, 110552. <https://doi.org/10.1016/j.msec.2019.110552> (2020).
- Zauska, L. *et al.* Thermosensitive drug delivery system SBA-15-PEI for controlled release of nonsteroidal anti-inflammatory drug diclofenac sodium salt: A comparative study. *Materials* **14**, 1880. <https://doi.org/10.3390/ma14081880> (2021).
- Stoddart, A. Predicting perfect pores. *Nat. Rev. Mater.* **5**, 331. <https://doi.org/10.1038/s41578-020-0200-6> (2020).
- Zhang, X. *et al.* Optimization of the pore structures of MOFs for record high hydrogen volumetric working capacity. *Adv. Mater.* **32**, 1907995. <https://doi.org/10.1002/adma.201907995> (2020).
- Seymour, L. United States Patent 5,360,743, issued 1 (1994).
- Thommes, M. *et al.* Physisorption of gases, with special reference to the evaluation of surface area and pore size distribution (IUPAC Technical Report). *Pure Appl. Chem.* **87**, 1051–1069. <https://doi.org/10.1515/pac-2014-1117> (2015).
- Chui, S. S. Y., Lo, S. M. F., Charmant, J. P. H., Orpen, A. G. & Williams, I. D. A chemically functionalizable nanoporous material [Cu₃(TMA)₂(H₂O)₃]_n. *Science* **283**, 1148–1150. <https://doi.org/10.1126/science.283.5405.1148> (1999).

37. Madalan, A. M., Maxim, C., Jurca, B., Avarvari, N. & Andruh, M. Constructing robust channel structures by packing metallacalixarenes: Reversible single-crystal-to-single-crystal dehydration. *J. Am. Chem. Soc.* **131**, 4586–4587. <https://doi.org/10.1021/ja900416e> (2009).
38. Konar, S. *et al.* A new porous 2D coordination polymer built by copper(II) and trimesic acid. *Inorg. Chim. Acta* **358**, 29–35. <https://doi.org/10.1016/j.ica.2004.08.010> (2005).
39. Gaikwad, S., Kim, Y., Gaikwad, R. & Han, S. Enhanced CO₂ capture capacity of amine-functionalized MOF-177 metal organic framework. *J. Environ. Chem. Eng.* **9**, 105523. <https://doi.org/10.1016/j.jece.2021.105523> (2021).
40. Flaig, R. W. *et al.* The chemistry of CO₂ capture in an amine-functionalized metal–organic framework under dry and humid conditions. *J. Am. Chem. Soc.* **139**, 12125–12128. <https://doi.org/10.1021/jacs.7b06382> (2017).
41. Xian, S., Wu, Y., Wu, J., Wang, X. & Xiao, J. Enhanced dynamic CO₂ adsorption capacity and CO₂/CH₄ selectivity on polyethyleneimine-impregnated UiO-66. *Ind. Eng. Chem. Res.* **54**, 11151–11158. <https://doi.org/10.1021/acs.iecr.5b03517> (2015).
42. Wang, N., Mundstock, A., Liu, Y., Huang, A. & Caro, J. Amine-modified Mg-MOF-74/CPO-27-Mg membrane with enhanced H₂/CO₂ separation. *Chem. Eng. Sci.* **124**, 27–36. <https://doi.org/10.1016/j.ces.2014.10.037> (2015).
43. Fan, H., Xia, H., Kong, C. & Chen, L. Synthesis of thin amine-functionalized MIL-53 membrane with high hydrogen permeability. *Int. J. Hydrog. Energy* **38**, 10795–10801. <https://doi.org/10.1016/j.ijhydene.2013.02.040> (2013).
44. Gadipelli, S. *et al.* Nanoconfinement and catalytic dehydrogenation of ammonia borane by magnesium–metal–organic–framework-74. *Chem. Eur. J.* **17**, 6043–6047. <https://doi.org/10.1002/chem.201100090> (2011).
45. Xia, L. & Liu, Q. Adsorption of H₂ on aluminum-based metal-organic frameworks: A computational study. *Comput. Mater. Sci.* **126**, 176–181. <https://doi.org/10.1016/j.commatsci.2016.09.039> (2017).
46. Sumida, K. *et al.* Hydrogen storage properties and neutron scattering studies of Mg₂(dobdc)—A metal–organic framework with open Mg²⁺ adsorption sites. *Chem. Commun.* **47**, 1157–1159. <https://doi.org/10.1039/c0cc03453c> (2011).

Author contributions

T.Z.: Conceptualization, Data curation, Formal analysis, Funding acquisition, Investigation, Project administration, Methodology, Supervision, Validation, Writing—original draft, Writing—review & editing; K.S.: Data curation, Formal analysis, Investigation, Methodology, Validation; R.S.: Data curation, Formal analysis, Investigation, Methodology, Validation; G.Z.: Data curation, Formal analysis, Investigation, Methodology, Validation, Writing—review & editing; S.P.N.: Data curation, Formal analysis, Investigation, Methodology, Validation; A.S.: Data curation, Formal analysis, Investigation, Methodology, Supervision, Validation, Writing—original draft, Writing—review & editing; M.A.: Conceptualization, Data curation, Formal analysis, Funding acquisition, Investigation, Project administration, Methodology, Supervision, Validation, Writing—original draft, Writing—review & editing. All authors read and approved the final manuscript.

Funding

This work was supported by the projects: VEGA-1/0865/21, KEGA-006UPJS-4/2021, APVV SK-CZ-RD-21-0068, LUASK22049 (INTER-EXCELLENCE II, MŠMT), VVGS-2022-2123, SGS16/PřF/2022, and TRIANGEL. The authors thank to Dr. Veronika Kucharova from the Institute of Experimental Physics, the Slovak Academy of Sciences for related TG/DTA measurements.

Competing interests

The authors declare no competing interests.

Additional information

Supplementary Information The online version contains supplementary material available at <https://doi.org/10.1038/s41598-022-22273-2>.

Correspondence and requests for materials should be addressed to M.A.

Reprints and permissions information is available at www.nature.com/reprints.

Publisher's note Springer Nature remains neutral with regard to jurisdictional claims in published maps and institutional affiliations.



Open Access This article is licensed under a Creative Commons Attribution 4.0 International License, which permits use, sharing, adaptation, distribution and reproduction in any medium or format, as long as you give appropriate credit to the original author(s) and the source, provide a link to the Creative Commons licence, and indicate if changes were made. The images or other third party material in this article are included in the article's Creative Commons licence, unless indicated otherwise in a credit line to the material. If material is not included in the article's Creative Commons licence and your intended use is not permitted by statutory regulation or exceeds the permitted use, you will need to obtain permission directly from the copyright holder. To view a copy of this licence, visit <http://creativecommons.org/licenses/by/4.0/>.

© The Author(s) 2022

Terms and Conditions

Springer Nature journal content, brought to you courtesy of Springer Nature Customer Service Center GmbH (“Springer Nature”).

Springer Nature supports a reasonable amount of sharing of research papers by authors, subscribers and authorised users (“Users”), for small-scale personal, non-commercial use provided that all copyright, trade and service marks and other proprietary notices are maintained. By accessing, sharing, receiving or otherwise using the Springer Nature journal content you agree to these terms of use (“Terms”). For these purposes, Springer Nature considers academic use (by researchers and students) to be non-commercial.

These Terms are supplementary and will apply in addition to any applicable website terms and conditions, a relevant site licence or a personal subscription. These Terms will prevail over any conflict or ambiguity with regards to the relevant terms, a site licence or a personal subscription (to the extent of the conflict or ambiguity only). For Creative Commons-licensed articles, the terms of the Creative Commons license used will apply.

We collect and use personal data to provide access to the Springer Nature journal content. We may also use these personal data internally within ResearchGate and Springer Nature and as agreed share it, in an anonymised way, for purposes of tracking, analysis and reporting. We will not otherwise disclose your personal data outside the ResearchGate or the Springer Nature group of companies unless we have your permission as detailed in the Privacy Policy.

While Users may use the Springer Nature journal content for small scale, personal non-commercial use, it is important to note that Users may not:

1. use such content for the purpose of providing other users with access on a regular or large scale basis or as a means to circumvent access control;
2. use such content where to do so would be considered a criminal or statutory offence in any jurisdiction, or gives rise to civil liability, or is otherwise unlawful;
3. falsely or misleadingly imply or suggest endorsement, approval, sponsorship, or association unless explicitly agreed to by Springer Nature in writing;
4. use bots or other automated methods to access the content or redirect messages
5. override any security feature or exclusionary protocol; or
6. share the content in order to create substitute for Springer Nature products or services or a systematic database of Springer Nature journal content.

In line with the restriction against commercial use, Springer Nature does not permit the creation of a product or service that creates revenue, royalties, rent or income from our content or its inclusion as part of a paid for service or for other commercial gain. Springer Nature journal content cannot be used for inter-library loans and librarians may not upload Springer Nature journal content on a large scale into their, or any other, institutional repository.

These terms of use are reviewed regularly and may be amended at any time. Springer Nature is not obligated to publish any information or content on this website and may remove it or features or functionality at our sole discretion, at any time with or without notice. Springer Nature may revoke this licence to you at any time and remove access to any copies of the Springer Nature journal content which have been saved.

To the fullest extent permitted by law, Springer Nature makes no warranties, representations or guarantees to Users, either express or implied with respect to the Springer nature journal content and all parties disclaim and waive any implied warranties or warranties imposed by law, including merchantability or fitness for any particular purpose.

Please note that these rights do not automatically extend to content, data or other material published by Springer Nature that may be licensed from third parties.

If you would like to use or distribute our Springer Nature journal content to a wider audience or on a regular basis or in any other manner not expressly permitted by these Terms, please contact Springer Nature at

onlineservice@springernature.com